

PATENT ABSTRACTS OF JAPAN

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(71) Applicant: TAJHEIYO KINZOKU KK

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WATANABE ATSUSHI KATO KANEYUKI

(54) POWDER AND METHOD FOR MANUFACTURING HIGH DENSITY SINTERED

(57) Abstract:

PROBLEM TO BE SOLVED: To stably manufacture a sintered parts of high sintering density by a pressing method by granulating powder small in mean grain size (ne powder) into granulated powder, setting the mean grain size to be approximately on the same level as that of the powder large in mean grain size (general powder), and mixing the granulated powder with the general powder to be formed and sintered.

SOLUTION: Raw ne powder and raw general powder are metallic powders formed of iron, iron alloy, nickel, etc. The ne powder is preferably 1-100 μ m in mean grain size. The ne powder is fed to a granulating device, a binder such as polyvinyl alcohol and polyvinyl pyrolidone is added thereto, and mixed therein to be granulated. A mixing granulating method is preferably of a rolling type in which a rotary pan, a rotary cylinder or a rotary cone is used. The binder is removed simultaneously with a lubricant in a degreasing process after the mixed powder is formed. A parts of the sintering density of >93% can be easily and inexpensively manufactured by the pressing method.

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WATANABE ATSUSHI KATO KANEYUKI

(54) POWDER AND METHOD FOR MANUFACTURING HIGH DENSITY SINTERED BODY

(57)Abstract:

PROBLEM TO BE SOLVED: To stably manufacture a sintered parts of high sintering density by a pressing method by granulating powder small in mean grain size (fine powder) into granulated powder, setting the mean grain size to be approximately on the same level as that of the powder large in mean grain size (general powder), and mixing the granulated powder with the general powder to be formed and sintered. SOLUTION: Raw fine powder and raw general powder are metallic powders formed of iron, iron alloy, nickel, etc. The fine powder is preferably 1−100 µm in mean grain size. The fine powder is fed to a granulating device, a binder such as polyvinyl alcohol and polyvinyl pyrolidone is added thereto, and mixed therein to be granulated. A mixing granulating method is preferably of a rolling type in which a rotary pan, a rotary cylinder or a rotary cone is used. The binder is removed simultaneously with a lubricant in a degreasing process after the mixed powder is formed. A parts of the sintering density of ≥93% can be easily and inexpensively manufactured by the pressing method.

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(54) 【発明の名称】 高密度焼結体製造用粉末および高密度焼結体の製造方法

(57)【要約】

【課題】 本発明は、高い焼結密度を有する焼結部品を プレス成形法により安定的に製造するための粉末及びそ の方法を堤供することを目的としている。

【解決手段】 平均粒径が小さい粉末を造粒して製造し た造粒粉末と、平均粒径が大きな粉末とを混合して得ら れた混合粉末、及びこの混合粉末を成形し、焼結するこ とを特徴とする高密度焼結体の製造方法。

【効果】

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【特許請求の範囲】

【請求項1】 平均粒径が小さい粉末を造粒して製造し た造粒粉末と、平均粒径が大きな粉末とを混合して、混 合粉末を構成することを特徴とする高密度焼結体製造用 粉末。

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【請求項2】 平均粒径が小さい粉末を造粒して製造し た造粒粉末と、平均粒径が大きな粉末とを混合して混合 粉末とし、この混合粉末を成形し、焼結することを特徴 とする高密度焼結体の製造方法。

【発明の詳細な説明】

[0001]

【発明の属する技術分野】本発明は、粉末冶金法による 焼結体用粉末及び焼結体の製造方法に関するものであ

[0002]

【従来の技術】粉末冶金法による焼結部品は部品形状の 制約がなく、また機械加工を行うことなく小型で複雑な 形状の部品を精度良く製造できるという特長がある。こ のため、粉末冶金法による焼結部品は、最近、機械、自 動車、家庭電気製品などに使用される部品として注目さ 20 れている。このような焼結部品は、例えばステンレス鋼 粉末を射出成形、あるいはプレス成形して所定形状の成 形体を成形し、これを焼結する方法で製造されている。 【0003】このような焼結部品の製造にあたっては、

最近、特に焼結密度を髙く保ち、さらに機械的強度を髙 いレベルに維持することが求められてきている。例え ば、自動車の排気ガス系統における部品に、ステンレス 鋼粉末を原料とする焼結部品が多用されるようになって きたが、このような部品には、燃焼ガスの漏出防止のた めに高い焼結密度を持ち、開放気孔をもたないこと、自 30 動車の振動による応力に耐える強度を持つこと、ガス等 による酸化や腐食に耐えること等が要求されている。

【0004】このうち、焼結密度については、真密度に 対する実測密度の比である相対密度が目安として用いら れ、少なくとも93%、好ましくは95%以上とするこ とが求められている。93%ないしは95%以上の相対 密度をもつ焼結部品であれば、開放気孔、すなわち焼結 部品の厚さ全体を貫く気孔、が存在しないと言われてい

【0005】このような高い焼結密度を有する焼結部品 40 を製造する方法がいくつか提案されている。例えば、特 開平7-316603号においては、「平均粒径が11 μ m以下の粉末で、該粉末中に含まれる 2 1 μ m以上の 粒径を有する粉末の存在率が17.5wt%未満である ことを特徴とする高密度及び高強度焼結体用粉末」を用 いることにより、「相対密度95%以上の焼結体が得ら れ、高い機械的特性を安定して得ることができる」こと を開示している。

[0006]

結密度の焼結部品を製造する方法として特開平7-31 6603号に開示された方法を用いようとすると、下記 の問題点を有している。すなわち、

- (1) 粉末の平均粒径が小さすぎるために、このよう な粉末を使用してプレス成形法によって焼結部品を製造 しようとすると、粉末の流動性が極端に低下し、成形装 置に粉末が流れ込まないため、正常な形状の焼結部品が 得られない。特開平7一316603号の実施例におい ても、射出成形法によってのみ部品を製造している。
- (2) 平均粒径の小さい粉末は一般的にコストが高い ため、このような粉末のみで焼結部品を製造しても経済 性をもたない。

【0007】本発明は上記のような実情に鑑みてなされ たもので、高い焼結密度を有する焼結部品をプレス成形 法により安定的に製造するための粉末及びその方法を堤 供することを目的としている。

【0008】従来、プレス成形法による焼結部品の製造 には、-100メッシュ品に代表されるような粉末が原 料として使用され、その相対密度は85%程度の低いも のしか得られなかったが、本発明は、プレス成形法によ り客易に、かつ安価に93%以上の焼結密度を有する焼 結部品を製造できる粉末及びその方法を提供することを 目的としている。

[0009]

【課題を解決するための手段】本件発明者らは、上記問 題を解決するために、種々検討を行った結果、「平均粒 径の小さい粉末」(以下、「微細粉末」と呼ぶ)を造粒 して造粒粉末とし、その造粒粉末の平均粒径を、平均粒 径の大きな粉末(以下、「一般粉末」と呼ぶ)とほぼ同 レベルにした後、この造粒粉末と一般粉末とを混合して 混合粉末とし、この混合粉末を成形し、焼結すれば、上 記の問題を解決して、高い焼結密度を有する焼結部品を 製造できることを見いだした。

【0010】ここで、本発明で原料となる粉末は微細粉 末、一般粉末ともに金属粉末であり、本発明は、鉄、鉄 合金(例えば、ステンレス鋼、珪素鋼、構造用鋼、工具 鋼、高速度鋼、アルニコ合金、センダスト合金、パーメ ンジュール合金、パーマロイ合金等)、ニッケル、ニッ ケル合金(パーマロイ等)、クロム、クロム合金、コバ ルト、コバルト合金、チタン、チタン合金等の金属粉末 に利用できるものである。

【0011】本発明において用いる微細粉末としては、 その平均粒径が1~100μmの金属粉末が好ましい。 平均粒径が 1 μ m以下の金属粉末を造粒しようとする と、バインダーの必要量が増加したり、またバインダー の除去に時間を要したりしてコスト負担が大きい。ま た、原料の金属粉末の平均粒径が100μmを越える と、この粉末で製造した造粒粉末を成形した焼結部品の 充填密度が低く、その後これを焼結しても密度は上昇せ 【発明が解決しようとする課題】しかしながら、高い焼 so ず、さらに寸法精度も悪くなる。従って、原料の微細粉 3

末の平均粒径の上限は100μmである。原料の微細粉 末の製造条件、経済性を考慮すると、好ましい原料の金 属粉末の平均粒径は2μm~60μmである。

【0012】次に、原料の微細粉末を造粒するには、微細粉末を造粒装置に供給し、バインダーを添加、混合して造粒する。造粒法としては混合造粒法、強制造粒法、熱利用造粒法のいずれを使用してもよいが、好ましくは、回転皿や回転円筒、回転円錐を使用する転動方式の混合造粒法を使用する。また、バインダーとしてはポリビニルアルコール、ポリビニルピロリドン、ワックス、寒天等の有機物や水と有機物の混合物等を使用する。

【0013】この造粒粉末を製造する時に添加するポリビニルアルコールのようなバインダーは特に除去工程を設ける必要はなく、混合粉末を成形した後に行う脱脂工程において、潤滑剤と同時に除去される。造粒工程において添加するバインダーと、成形工程において添加する潤滑剤の性質が異なり、潤滑剤に比較してバインダーが除去しにくい時には、脱脂時間を延長したり、あるいは脱脂温度を高める等、脱脂条件を調整することが必要である。

【0014】また、造粒粉末と一般粉末とを混合して混合粉末とし、この混合粉末を成形し、焼結する時の焼結

条件としては、通常使用されている水素雰囲気下での焼結、窒素、アルゴン等の不活性雰囲気下での焼結、減 圧、真空又は酸化性雰囲気下での焼結等、公知の焼結条 件でよい。

[0015]

【実施例】以下、本発明の要旨を実施例によりさらに詳細に説明する。

[0016]

【実施例1】微細粉末として、表1に示す粒径分布を有する、水アトマイズ法により得られたステンレス鋼(SUS316L)粉末50kgを転動流動造粒装置(株式会杜パウレック製)に供給し、回転ディスク部を回転させながらバインダーとしてメチルアルコールを溶媒とするPVP(ポリビニルピロリドン)5%溶液を連続的に供給して造粒した。この際の空気供給量は約8.5m³/hr、給気温度は70℃、回転ディスク部の回転数は約205rpmで、造粒粉末の温度は約35℃、バインダーの供給量は約110g/分、造粒時間は180分、乾燥時間は約10分とした。

20 [0017]

【表1】

	粒径	+ 40	- 40	- 30	- 20	- 10	平均粒径
	(µm)		~+ 30	~+ 20	~+ 10		(µm)
	質量比	2.9	4.3	13.1	29.2	50.5	9.88
ļ	(%)						

この造粒によって、表2に示す粒径分布の造粒粉末を得

【0018】 【表2】

た。

粒径		- 177 ~+ 149				
(μm) 質量比	0.3	5.4	15.0	14.6	22.1	42.6
(%)						

次に表2の造粒粉末と、一般粉末として表3に示す粒径 分布を有する水アトマイズ法により得られたステンレス 鋼(SUS316L)粉末を混合し、混合粉末を作成し た(この一般粉末は、通常、いわゆる-100メッシュ に際して、造粒粉末と一般粉末の混合比率を種々に変化 させた。

[0019]

【表3】

品と呼ばれている粉末である。)。この混合粉末の作成

~ o / o ·						
粒径	+ 177	- 177	- 149	- 105	- 74	- 44
(μm)		~+ 149	~+ 105	~ + 74	~+ 44	
質量比	0.0	0.4	8.1	19.4	33.0	39.1
(%)						

この混合粉末に潤滑剤として市販のACRAWAX(ロンザ・ジャパン株式会社製)を質量比で0.5%添加した後、プレス機を使用して成型圧力6~8t/cm²で試験片を作り、水紫雰囲気の焼結炉で1200℃×1時間焼結した後、焼結密度を測定した。

【0020】その結果は表4に示すように、一般粉末と 50

造粒粉末の混合比、成形圧力をかえることにより、相対 密度93%以上の焼結部品を製造できる可能性のあることが分かる。

[0021]

【表4】

5

混合比率	相対密度(%)			
(質量比)	成形	E力(t/c	m")	
一般粉末 造粒粉末	6	7	8	
100部 0部	84. 4	86. 5	88. 3	
75部 25部	87. 5	89. 6	90.4	
50部 50部	89. 7	91.1	91.9	
25部 75部	92. 2	93.0	93.6	
0部 100部	93.8	94.7	95.6	

[0022]

【実施例2】次に、実施例1と同一条件で成形した試験 片を、焼結温度の影響を調査するため、水素雰囲気の焼 結炉で1300℃×1時間の条件で焼結した後、焼結密 度を測定した。 【0023】その結果を表5に示すように、焼結温度を 高めることにより、相対密度93%以上の焼結部品を製 造できる範囲が広がってくることがわかる。

[0024]

【表5】

	混合比率		対密度(%))		
(質)	胜)	成形	成形圧力(t / c m²)			
一般粉末	造粒粉末	6	7	8		
100部	0部	86. 7	88. 7	90.8		
75部	25部	90.0	92.1	93.0		
50部	50部	93.0	93.6	94.2		
25部	75部	94.6	95.7	96.0		
0部	100部	96.0	96. 2	97. 2		

30

[0025]

【実施例3】次に実施例1と同じ微細粉末を用いて、実施例1と同様の方法により、実施例1の場合よりも粒径の小さい表6の造粒粉末を得た。

[0026]

【表6】

.01				
粒 径	+ 74	- 74	- 63	- 44
(μm)		∼ +63	~+ 44	
質量比	2.8	9.3	23.6	64.3
(%)				

次に表6の造粒粉末と、一般粉末として表7に示す粒径分市を有する水アトマイズ法により得られたステンレス鋼(SUS316L)粉末を混合し、混合粉末を作成した。

[0027]

【表7】

拉径	+ 74	- 74 ~+ 63	- 63 ~+ 44	
質量比 (%)	1.1	12.8	25.7	60.4

この混合粉末に潤滑剤として市販のACRAWAX(ロンザ・ジャパン株式会社製)を質量比で0.5%添加した後、プレス機を使用して成型圧力 $6\sim8$ t / c m^2 で試験片を作り、水素雰囲気の焼結炉で1300 $\mathbb{C}\times1$ 時間焼結した後、焼結密度を測定した。

【0028】その結果は表8に示すように、微細粉末、一般粉末及び造粒粉末の粒径をさらに細かいものに揃えることにより、安定して93%以上の相対密度を持つ焼結部品を製造できることが確認できた。

40 [0029]

【表8】

	混合比率		相対密度(%)				
實)	胜)	咸	ΙΈ	E力(t	/ c	m³)	
一般粉末	追粒粉末	6		7		8	
100部	0部	90.	5	92.	3	92.	8
75部	25部	93.	5	93.	9	95.	1
50部	50部	95.	4	95.	6	95.	8
25部	75部	96.	3	96.	6	97.	0
0部	100部	97.	3	97.	2	97.	3

[0030]

【実施例4】同様の実験を水アトマイズ法により得られ たステンレス鋼(SUS410L)について行った。微 細粉末、造粒粉末、一般粉末の粒径分布は実施例1の場 合とほぼ同様のものを用いた。また、造粒条件、焼結条 件も実施例1と同様とした。

【0031】その結果は表9に示すように、同じステン レス鋼でもSUS410Lの場合は、焼結温度1200 ℃でも、安定して93%以上の相対密度を有する焼結体 を得ることができることがわかる。

[0032]

【表9】

			Lacul				
	混合比率		相対密度(%)				
(質)	批)	成形E	E力(t/c	m*)			
一般粉末	追拉粉末	6	7	8			
100部	0部	88.6	89. 9	90.9			
75部	25部	93.0	93.2	93.9			
50部	50部	95. 1	95.2	95.5			
25部	75部	96.4	96.1	96.6			
0部	100部	96.9	96.9	97.4			

以上のように、微細粉末・造粒粉末・一般粉末の粒径、 造粒粉末と一般粉末の混合比率、成形圧力、焼結温度等 の条件を適宜選択することにより、93%以上の焼結密 造粒粉末と一般粉末の混合比率の選択は特に重要である が、93%以上の相対密度を確保した上で、さらに必要 な相対密度、コスト等の条件を勘案して、最適な混合比 率を選択する必要がある。

[0033]

【発明の効果】本発明により平均粒径の小さい粉末を造 度を有する焼結部品を安定して製造することができる。 30 粒して造粒粉末とし、この造粒粉末と平均粒怪の大きい 粉末の混合粉末を製造して、この混合粉末を成形・焼結 すれば、高い焼結密度を有する焼結部品を製造すること ができ、その効果は極めて大きい。

フロントページの続き

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(54) [Title of the Invention] A powder and a method for manufacturing a high density sintered body

(57) [Abstract]

[Problem] The present invention intends to provide a powder and a method for stably manufacturing sintered parts of high sintering density by pressing.

[Means for Solving] A mixed powder obtained via blending a granulated powder which is manufactured by granulating a powder small in average granular diameter with a powder large in average granular diameter, and a manufacturing method of a high density sintered body, comprised of molding and sintering said mixed powder.

[Effect]

[Scope of Patent Claims]

[Claim 1] A powder for manufacturing a high density sintered body, said powder comprising a mixed powder via blending a granulated powder which is manufactured by granulating a powder small in average granular diameter with one large in average granular diameter.

[Claim 2] A manufacturing method of a high density sintered body, said manufacturing method comprising: blending the granulated powder which is manufactured by granulating the powder small in average granular diameter with one large in average granular diameter to give a mixed powder, then molding and sintering said mixed powder.

[Detailed Description of the Invention]

[0001]

[Technical Field of the Invention] The present invention relates to a powder and a method for manufacturing a sintered body via a powder metallurgy process.

[0002]

[Prior Art] Parts sintered via a powder metallurgy process are characterized by having no limitation with regard to shape and can be accurately manufactured as small complicated parts without the need of machine processing. Thus, parts sintered via a powder metallurgy process have recently come to be known as parts which can be used for machines, automobiles, and electric home appliances. These sintered parts are manufactured, for example, by injection molding or pressing of stainless steel powder to form and sinter a specific shape of compacts. [0003]

For manufacturing such sintered parts, recently it has been required to maintain, in particular, the sintering density as well as the mechanical strength at a high level. For example, sintered parts comprised of stainless steel powder as raw material have come to be commonly used in parts for automobile exhaust gas systems. These parts are required to have high sintering density for preventing the leakage of fuel gas, no open holes, strength to resist the stress resulting from vibration of the automobile, and resistance to oxidization and corrosion by gas and the like. [0004]

With these parts, the relative density, which is a ratio of the actual measured density to the true density, has been used as a standard for the sintering density. It is required to be at least 93% and preferably over 95%. It is said that sintered parts with relative densities of 93% or over 95% do not have opening holes, that is, holes penetrating the thickness of the sintered parts. [0005]

Several methods for manufacturing these sintered parts result in a high sintering density. For example, Japanese unexamined patent application publication No. 1995-316603 disclosed that "a sintered body having a relative density of 95% or greater can be obtained and high-mechanical properties can be stably achieved" by using "a powder for a high density and high strength sintered body that is characterized by having an average granular diameter of 11µm or less and in which granules of 21µm or greater make up less than 17.5 wt% of the total powder content." [0006]

[Problem to be Solved by the Invention] However, when attempting to use the method disclosed in Japanese unexamined patent application publication No. 1995-316603 as a manufacturing method of high density sintered parts, the following problems occur. Briefly, (1) since the average granular diameter of the powder is too small, when manufacturing sintered parts by a pressing method using such powder, the powder is not fed into the molding device. Therefore, sintered parts of normal shape cannot be obtained. In the embodiment described in Japanese unexamined patent application publication No. 1995-316603, the parts are manufactured merely by an injection molding method; and (2) since a powder small in average granular diameter is generally expensive, it is uneconomical to manufacture sintered parts with such powder alone. [0007]

The present invention was developed in consideration of said circumstances. It intends to provide a powder and a method for stably manufacturing sintered parts of high sintering density by a pressing method.

[8000]

Traditionally, for manufacturing sintered parts by a pressing method, powder represented by -100 mesh product has been used as raw material, which has only produced parts with a low relative density of 85%. However, the present invention intends to provide a powder and a method that enables the simple and inexpensive production of sintered parts with a sintering density of 93% or greater by a pressing method.

[0009]

[Means for Solving the Problem] The present inventors have conducted various studies in order to solve said problems. As a result, it was found that said problems can be solved and sintered parts of high sintering density produced by granulating a "powder small in average granular diameter" (hereinafter referred to as "fine powder") to give a granulated powder, forming said granulated powder to have an average granular diameter substantially the same as that of a powder large in average granular diameter (hereinafter referred to as "general powder"), then blending said granulated powder and said general powder to give a mixed powder, and forming/sintering said mixed powder.

[0010]

Wherein, raw fine powder and raw general powder of the present invention are metallic powders. The present invention can be utilized with metallic powders including iron, iron alloy (for example, stainless steel, silicon steel, structural steel, tool steel, high-speed steel, alnico alloy, Sendust alloy, Permendur alloy, Permalloy alloy, etc.), nickel, nickel alloy (Permalloy, etc.), chromium, chromium alloy, cobalt, cobalt alloy, titanium, titanium alloy and the like.

[0011]

The fine powder used in the present invention is preferably a metallic powder with an average granular diameter of 1-100 μ m. Attempting to granulate metallic powder with an average granular diameter of 1 μ m or less increases the cost due to the increase in the required volume of binder and the additional time for removing the binder. In addition, when the average granular diameter of raw metallic powder exceeds 100 μ m, the sintered parts formed with the granulated powder comprised of said powder have low filling density. The density does not increase after sintering the parts, and further, the dimension accuracy deteriorates. Accordingly, the upper limit of the average granular diameter of raw fine powder is determined to be 100 μ m. Taking into consideration the manufacturing conditions and economic efficiency of raw fine powder, the preferred average granular diameter of raw metallic powder falls between 2μ m-60 μ m.

Next, in order to granulate raw fine powder, the fine powder is fed into a granulating device, and a binder is added thereto and mixed therein to be granulated. Although any of mixing granulation, forced granulation or heat granulation can be used as a granulating method, rolling mixing granulation that uses a rotating disk, rotating cylinder, and rotating cone is preferred. In addition, a binder, such as organic materials including polyvinyl alcohol, polyvinyl pyrolidone, wax, and agar along with a mixture of water and said organic material, is used.

[0013]

Regarding the binder, such as polyvinyl alcohol, in order to be added in the manufacturing process of the granulated powder, it is not particularly necessary to include a removing process. The binder is removed simultaneously with a lubricant in a degreasing process after the mixed powder is formed. If it is difficult to remove the binder in comparison with the lubricant due to the difference in the nature of the binder to be added in the granulating process and that of the lubricant to be added in the forming process, it is necessary to adjust the degreasing conditions, such as by extending the degreasing time or the degreasing temperature. [0014]

Regarding the sintering conditions for blending the granulated powder and the general powder to give the mixed powder, and then forming and sintering the mixed powder, known sintering conditions can be applied; for example, sintering under a hydrogen atmosphere that is typically

used, sintering under an inert atmosphere, such as nitrogen, argon, etc., and sintering under a reduced pressure, vacuum, and oxidizing atmosphere.

[0015]

[Embodiment] The summary of the present invention will be explained in detail below. [0016]

[Embodiment 1] As fine powder, 50kg of stainless steel (SUS316L) powder obtained by a water-atomizing method with the granular diameter distribution shown in Table 1 was fed into a rolling current granulating device (manufactured by Powrex Corp.). While rotating the disk part, 5% PVP (polyvinyl pyrolidone) solution containing methyl alcohol as solvent was continuously supplied as a binder for the granulation. The conditions were determined to be as follows: air supply volume: approx. 8.5m³/hr, charge air temperature: 70°C, revolution of the rotating disk part: approx. 205rpm, temperature of the granulated powder: approx. 35°C, supply volume of the binder: approx. 110g/min, granulating time: 180 minutes, and drying time: approx. 10 minutes. [0017]

[Table 1]

[Table 1]						
Granular	+40	-40	-30	-20	-10	Average
diameter		~+30	~+20	~+10		granular
(µm)						diameter
						(µm)
Mass ratio (%)	2.9	4.3	13.1	29.2	50.5	9.88

From the granulating process, granulated powder having the granular diameter distribution shown in Table 2 was obtained.

[0018] [Table 2]

<u> </u>						
Granular	+177	-177	-149	-105	-74	-44
diameter		~+149	~+105	~+74	~+44	
(µm)						
Mass ratio	0.3	5.4	15.0	14.6	22.1	42.6
(%)						

Next, for the general powder, the granulated powder shown in Table 2 was mixed with a stainless steel (SUS316L) powder obtained by a water-atomizing method having the granular diameter distribution shown in Table 3 to give the mixed powder (said general powder is typically a so called powder -100 mesh product). When producing the mixed powder, the blending ratio of the granulated powder and the general powder was changed in various ways. [0019]

[Table 3]

[14010 5]						
Granular	+177	-177	-149	-105	-74	-44
diameter (μm)	:	~+149	~+105	~+74	~+44	
Mass ratio (%)	0.0	0.4	8.1	19.4	33.0	39.1

A sample was produced by adding commercial ACRAWAX (manufactured by LONZA Japan) as a lubricant at 0.5% by weight to the mixed powder and pressing at 6-8t/cm² of molding pressure using a pressing machine. After the sample was sintered at 1200°C for 1 hour in a sintering furnace under a hydrogen atmosphere, the sintering density was measured. [0020]

From the results shown in Table 4, it was found that sintered parts having a relative density of 93% or greater can be manufactured by changing the blending ratio of the general powder and the granulated powder as well as the molding pressure.

[0021]

[Table 4]

Blendi	ng ratio	Relative density (%)			
(Mass	s ratio)	Molding pressure (t/cm ²)			
General	Granulated	6	7	8	
powder	powder				
100 unit	0 unit	84.4	86.5	88.3	
75 unit	25 unit	87.5	89.6	90.4	
50 unit	50 unit	89.7	91.1	91.9	
25 unit	75 unit	92.2	93.0	93.6	
0 unit	100 unit	93.8	94.7	95.6	

[0022]

[Embodiment 2] Next, in order to investigate the influence of the sintering temperature on the sample formed under the same conditions as in embodiment 1, the sample was sintered at 1300°C for 1 hour under a hydrogen atmosphere.

[0023]

From the results shown in Table 5, it was found that the range in which sintered parts having a relative density of 93% or greater can be manufactured can be expanded by raising the sintering temperature.

[0024]

[Table 5]

Blendi	ng ratio	Relative density (%)			
(Mass	s ratio)	Moldin	Molding pressure (t/cm ²)		
General	Granulated	6	7	8	
powder	powder				
100 unit	0 unit	86.7	88.7	90.8	
75 unit	25 unit	90.0	92.1	93.0	
50 unit	50 unit	93.0	93.6	94.2	
25 unit	75 unit	94.6	95.7	96.0	
0 unit	100 unit	96.0	96.2	97.2	

[0025]

[Embodiment 3] Using the same fine powder as in embodiment 1, by the same method as in embodiment 1, the granulated powder, shown in Table 6, of a smaller granular diameter than that of embodiment 1 was obtained.

[0026]

[Table 6]

Granular	+74	-74	-63	-44
diameter		~+63	~+44	

(µm)				
Mass	2.8	9.3	23.6	64.3
ratio (%)				

Next, a mixed powder was produced by blending the granulated powder shown in Table 6 and stainless steel (SUS316L) powder as the general powder obtained by a water-atomizing method with a granular diameter distribution as shown in Table 7. [0027]

[Table 7]

[rabic /]				
Granular	+74	-74	-63	-44
diameter		~+63	~+44	
(µm)				
Mass	1.1	12.8	25.7	60.4
ratio				
(%)				

A sample was produced by adding commercial ACRAWAX (manufactured by LONZA Japan) as a lubricant at 0.5% by weight to the mixed powder and pressing at 6-8t/cm² of molding pressure using a pressing machine. After the sample was sintered at 1300°C for 1 hour in a sintering furnace under a hydrogen atmosphere, the sintering density was measured. [0028]

From the results shown in Table 8, it was found that sintered parts stably having a relative density of over 93% can be manufactured by creating a uniform finer granular diameter of fine powder, general powder, and granulated powder.

[0029]

[Table 8]

Blendi	ng ratio	Relative density (%)					
(Mass	s ratio)	Molding pressure (t/cm ²)					
General	Granulated	6	7	8			
powder	powder						
100 unit	0 unit	90.5	92.3	92.8			
75 unit	25 unit	93.5	93.9	95.1			
50 unit	50 unit	95.4	95.6	95.8			
25 unit	75 unit	96.3	96.6	97.0			
0 unit	100 unit	97.3	97.2	97.3			

[0030]

[Embodiment 4] The same experiment was conducted for stainless steel (SUS410L) obtained by a water-atomizing method. A granular diameter distribution of fine powder, granulated powder, and general powder substantially the same as that of embodiment 1 was used. In addition, the granulating and sintering conditions applied to embodiment 1 were employed. [0031]

From the results shown in Table 9, in the case of using SUS410L from among stainless steels, it was found that a sintered body stably having a relative density of 93% or greater can be obtained even at a sintering temperature of 1200 °C. [0032]

[Table 9]

Blendi	ng ratio	Relative density (%)			
(Mass	(Mass ratio)		Molding pressure (t/cm ²)		
General	Granulated	6	7	8	
powder	powder				
100 unit	0 unit	88.6	89.9	90.9	
75 unit	25 unit	93.0	93.2	93.9	
50 unit	50 unit	95.1	95.2	95.5	
25 unit	75 unit	96.4	96.1	96.6	
0 unit	100 unit	96.9	96.9	97.4	

As shown above, by appropriately selecting the granular diameter of fine powder/granulated powder/general powder, the blending ratio of the granulated powder and the general powder, the molding pressure, the sintering temperature and the like, sintered parts having a sintering density of 93% or greater can be stably manufactured. In particular, selecting the blending ratio of the granulated powder and the general powder is important. Moreover, it is necessary to select an optimal blending ratio after ensuring a relative density of 93% or greater and take into consideration the conditions, such as required relative density, cost and the like. [0033]

[Effect of the Invention] According to the present invention, sintered parts of high sintering density can be manufactured by granulating powder small in average granular diameter to give a granulated powder, blending said granulated powder and a powder large in average granular diameter to produce a mixed powder, and molding/sintering said mixed powder, and the effect of the invention is significant.

Continued from the front page (72) Inventor: Kaneyuki KATO

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